# THE EFFECT OF REINFORCEMENT CONTENT FATIGUE CRACK GROWTH OF AL-SIC<sub>P</sub> METAL - MATRIX COMPOSITES

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# ABSTRACT

The purpose of this paper is to attempt to establish correlation between the varying contents of Mg and SiC<sub>p</sub>, and the fatigue crack resistance of the composite. The casted Al alloy was used as matrix, with 3, 5 and 7 % of Mg. Contents of the reinforcement, SiC particles were 2, 5 and 10 %. The size of the particle was 30  $\mu$ m. The results of the mechanical properties, crack resistance and fatigue tests, performed for all mentioned combinations of Al matrix alloy and percentage of reinforcement, are presented. The obtained results of tensile test, and crack growth rate da/dN vs.  $\Delta K$  is discussed and compared for each case of varying content of Mg and different content of SiC<sub>p</sub>.

Keywords: Young's modulus, strain to fracture, threshold value, Paris equation parameters

# 1. INTRODUCTION

In recent years  $Al-SiC_p$  metal matrix composites (MMC) have been introduced because of their superior mechanical properties such as combination of high stiffness, high strength and good tribological properties with low density and potentially low unit cost. It is well known that there are several problems regarding fabrication of this class of composites due to wettability and dispersion of SiC particles with molten matrix. Also, a variety of factors influence the microstructure and properties of cast  $Al-SiC_p$  composites [1-4]. This paper reports the influence of varying Mg and  $SiC_p$  content on fatigue crack growth.

# 2. MATERIAL DESCRIPTION

The matrix alloy used in this research is the Al with different content of the Mg. The chemical composition of the used matrix alloy is presented on Table 1. The composition of the reinforcement is presented on Table 2.

The size of the particle was 30  $\mu$ m. The composite was obtained by adding SiC particles into the melted matrix material, kept at certain temperature (proprietary information). The composite was

initially cast, by vortex method, in ingot form. The ingots were die cast into blocks, from which specimens used in this research were made. Stirring of mixture was applied during the casting to avoid particle clustering. Post heat treatment was not applied on casted ingots [5,6].

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	Matrix	Mg	Mn	Si	Fe	Ti	Zn	Cu	Al
	AlMg3	2,90	0,30	0,050	0,18	0,055	0,10	-	rest
	AlMg5	4,95	0,35	0,045	0,12	0,035	0,12	-	rest
	AlMg7	7,05	0,63	0,081	0,23	0,051	0,030	0,001	rest

Table 1. Composition of the matrix alloy, mass % [5,6]

Table 2. Composition of the reinforcement, mass % [5,6]

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С	Si+SiO <sub>2</sub>	TiO <sub>2</sub>	FeO <sub>3</sub>	$Al_2O_3$	CaO	MgO	SiC
0,12	1,88	0,14	0,50	0,14	0,13	0,04	96,65

Chosen contents of reinforcement are unusual, that is, too low to produce the significant strengthening. Basic idea was to produce inexpensive MMC with good resistance to fatigue crack growth.

# 3. EXPERIMENTAL PROCEDURES

The following experiments were performed on specimens made of the MMC:

- monotonic tensile loading to determine the material elasticity modulus, E, ultimate tensile strength,  $S_{ut}$  and strain to fracture,  $\epsilon_f$ ;
- fatigue crack growth testing.

Rectangular die cast blanks, 65 by 160 by 15 mm were used for all specimens. All tests were performed at room temperature (20°C on average). The dimensions of tensile specimens and load rate were chosen in accordance with ASTM E8-06 [7]. Results of tensile testing are presented on Fig. 1 to Fig 3.



Figure 1. Young's modulus vs. Vol. % of SiC<sub>p</sub>

Figure 2. The ultimate tensile strength vs. volume percent of  $SiC_n$ 

The fatigue crack growth tests at room temperature were performed under wave form (three points bending) constant amplitude load control on Charpy specimens. All tests were performed under the

same ratio of minimal and maximal load, R=0,1 [8]. Accuracy of acquisition of the average load and amplitude was  $\pm$ 5Ncm and frequencies were in range of 175-195 Hz. Results of da/dN vs.  $\Delta$ K are presented on Fig. 4 to Fig 6. Paris equation parameters and threshold values, C, n and  $\Delta$ K<sub>th</sub> obtained from da/dN vs.  $\Delta$ K curves are presented in Table 3.

MMC	R	С	n	$\Delta K_{th}$ , MPa m <sup>1/2</sup>
AlMg3 - 2% SiC	0,1	3,92·10 <sup>-15</sup>	9,301	3,81
AlMg3 - 5% SiC	0,1	$7,28 \cdot 10^{-15}$	8,861	3,95
AlMg3 - 10% SiC	0,1	$5,46 \cdot 10^{-16}$	10,845	4,05
AlMg5 - 2% SiC	0,1	$5,18 \cdot 10^{-15}$	8,027	4,05
AlMg5 - 5% SiC	0,1	$7,45 \cdot 10^{-14}$	7,065	4,10
AlMg5 - 10% SiC	0,1	6,49·10 <sup>-15</sup>	8,181	4,13
AlMg7 - 2% SiC	0,1	7,26.10-17	12,421	3,95
AlMg7 - 5% SiC	0,1	5,69.10-16	10,679	3,95
AlMg7 - 10% SiC	0,1	6,28·10 <sup>-16</sup>	10,426	4,20

Table 3. Results of values for  $\Delta K_{th}$ , C and n, for all combinations of matrix alloy and reinforcement



Figure 3. Strain to failure,  $\varepsilon_{f}$  vs. volume percent of SiC<sub>n</sub> reinforcement AIMg3 matrix alloy

#### 4. **DISCCUSION**

In Fig. 1, Fig 2 and Fig 3, observed scattering of all data can be explained with noted porosity of the obtained casted blocks. This scattering is especially visible for the strain to failure data, Fig 3. From the monotonic tensile load tests, the conclusions can be drawn that the best results were with AlMg5-10%SiC<sub>p</sub> combination [8].

Fig.4 to Fig 6. and Table 3. show that the growth rate of fatigue cracks is the greatest with AlMg5-10%SiC<sub>p</sub>. Scattering and deflection of curves, in region II, from Fig.6 to Fig.8 can be explained by non-uniformly distribution of reinforcement in all matrix alloys. With cracks propagation through region poor with reinforcement, crack closure effect was observed. Same was confirmed by fractography analysis. For the binary matrix system, Al-Mg, it is known that with increasing Mg content, portion of  $\beta$  phase, after 5% of Mg, significantly expands presence in the matrix. This is

confirmed by EDX analysis. The significant influence, during SEM analysis, of the  $\beta$  phase as crack initiation site is not observed. Virtually all crack initiation sites were on SiC-matrix interface (5,6).



Figure 5. Relation between da/dN vs.  $\Delta K$  for AlMg5 matrix alloy



Figure 6. Relation between da/dN vs.  $\Delta K$  for AlMg7 matrix alloy

# 5. CONCLUSIONS

The results of presented tests indicate that:

- Porosity and non-uniformity of  $SiC_p$  produced a scattering of the values of Young modulus, ultimate tensile strength and strain to failure.
- The  $\Delta K_{th}$  values are not significantly influenced with Mg content and portion of SiC<sub>p</sub>.
- Crack closure effects were pronounced, for all combinations of MMC, especially in Region II, due to non-uniformity of SiC particles.
- Fatigue crack growth rate is the lowest for AlMg5%-10SiC<sub>p</sub> combination, for presented route of producing MMC.

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